

Tunneling Magnetoresistance in a Two-Layered $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ Polycrystal

Tie-Jun Zhou, Z. Yu, and Y. W. Du

Department of Physics and National Laboratory of Solid State Microstructures, P. O. Box 2123, Nanjing University, Nanjing 210093, China

Reprint requests to T. J. Z.; E-mail: scmb@netra.nju.cn

Z. Naturforsch. **54 a**, 695–698 (1999); received August 20, 1999

By the sol-gel method, a polycrystal bulk of two-layered $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ with $x = 0.2$ was successfully synthesized. A high sensitive low-field TMR was found at low temperature.

PACS numbers: 75.30.Kz, 72.15.Gd

1. Introduction

The discovery of giant magnetoresistance (GMR) in magnetic multilayer films [1 - 2] and granular solids [3 - 4] has triggered considerable interest in relating the spin-dependent transport phenomena from the viewpoint of both the underlying physics and their immediate application to magnetic storage and sensor technology. A number of the GMR structures have so far been studied, such as magnetic multilayers with antiferromagnetic coupling, spin valves [5], granular solids, and tunneling structures [6]. Of all the GMR structures, the tunneling ones are currently investigated extensively because of their large magnetoresistance (MR) ratio in a very low field [7 - 9]. The MR ratio in tunneling structures is dependent on the spin polarization and the relative orientation of the magnetizations in the magnetic layers. Recently, a large tunneling magnetoresistance (TMR) as high as 83% was observed at low fields of only tens of Oe in an epitaxial tunnel junction with the form $\text{La}_{0.67}\text{Sr}_{0.33}\text{MnO}_3/\text{SrTiO}_3/\text{La}_{0.67}\text{Sr}_{0.33}\text{MnO}_3$, and this high MR value had been attributed to the almost full spin polarization in the manganite layers [10]. Also, a large low-field MR has been examined in polycrystalline manganites (TMR in granular solids) [11].

In fact, there exist natural tunneling structures, and one of them is layered perovskite manganite. This system intrinsically includes magnetic multilayers in its crystal structure (Figure 1). The simplest view

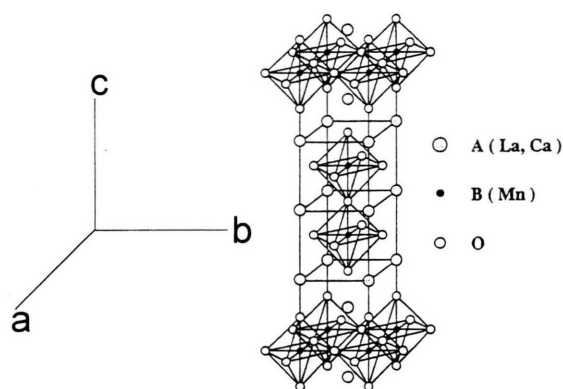


Fig. 1. Crystal structure of $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ with the $\text{Sr}_3\text{Ti}_2\text{O}_7$ -type structure.

of layered manganite is a stack of ferromagnetic-metallic sheets consisting of MnO_2 bilayers which are respectively separated by the $(\text{La}, \text{M})_2\text{O}_2$ layers which work as nonmagnetic insulating layers. That is to say the layered manganite crystal forms a naturally infinite array of ferromagnetic-metal / insulating / ferromagnetic-metal junctions. In this simple view, the nearly fully spin-polarized electrons' tunneling between the two MnO_2 bilayers should give a large low-field TMR. Very recently it was reported the TMR in two-layered perovskite single crystals can attain as large as 240 % in low field at low temperature and this value can be drastically enhanced up to ~4000% by applying a pressure of ~10 kbar at

0932-0784 / 99 / 1200-0695 \$ 06.00 © Verlag der Zeitschrift für Naturforschung, Tübingen · www.znaturforsch.com



Dieses Werk wurde im Jahr 2013 vom Verlag Zeitschrift für Naturforschung in Zusammenarbeit mit der Max-Planck-Gesellschaft zur Förderung der Wissenschaften e.V. digitalisiert und unter folgender Lizenz veröffentlicht: Creative Commons Namensnennung-Keine Bearbeitung 3.0 Deutschland Lizenz.

Zum 01.01.2015 ist eine Anpassung der Lizenzbedingungen (Entfall der Creative Commons Lizenzbedingung „Keine Bearbeitung“) beabsichtigt, um eine Nachnutzung auch im Rahmen zukünftiger wissenschaftlicher Nutzungsformen zu ermöglichen.

This work has been digitalized and published in 2013 by Verlag Zeitschrift für Naturforschung in cooperation with the Max Planck Society for the Advancement of Science under a Creative Commons Attribution-NoDerivs 3.0 Germany License.

On 01.01.2015 it is planned to change the License Conditions (the removal of the Creative Commons License condition “no derivative works”). This is to allow reuse in the area of future scientific usage.

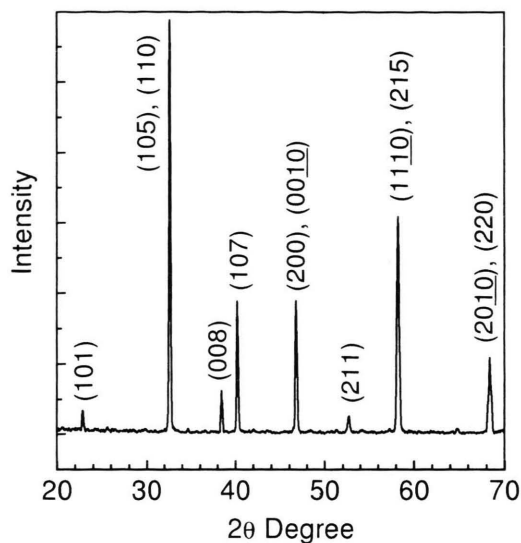


Fig. 2. The XRD pattern for the $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ with $x = 0.2$.

4.2 K [12, 13]. All of the past studies on layered perovskites, to the best knowledge of us, were focused on single crystals and films, and not on polycrystal bulks. In this paper, a polycrystal bulk of two-layered $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ with $x = 0.2$ was successfully synthesized, and a high-sensitive low-field TMR was discovered at low temperature.

The polycrystalline samples of $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ with $x = 0.2$ were prepared by the sol-gel method in order to obtain well-mixed reagents. Stoichiometric amounts of La_2O_3 , CaCO_3 and $\text{Mn}(\text{NO}_3)_2$ were dissolved in concentrated nitric acid. Afterwards, citric acid was added, the solution was heated at around 350 K for over two days, giving a gel, followed by decomposition at 470 K and calcination in air at 1470 K for 10 hours. The calcined material was ground and pressed into round disks. And final sintering process step was carried out at 1670 K in air for 30 hours, followed by cooling of the furnace. X-ray diffraction (XRD) and Scanning Electron Microscopy (SEM) were used to characterize the structure of the sample. The electrical resistance and MR were measured as function of temperature and applied field by a standard four-point technique. The MR ratio is defined as $\Delta R/R_0 = (R_H - R_0)/R_0$, where R_H and R_0 are the resistance at applied field H and zero-field resistance, respectively.

The XRD pattern for the $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ with $x = 0.2$ is presented in Figure 2. The diffraction

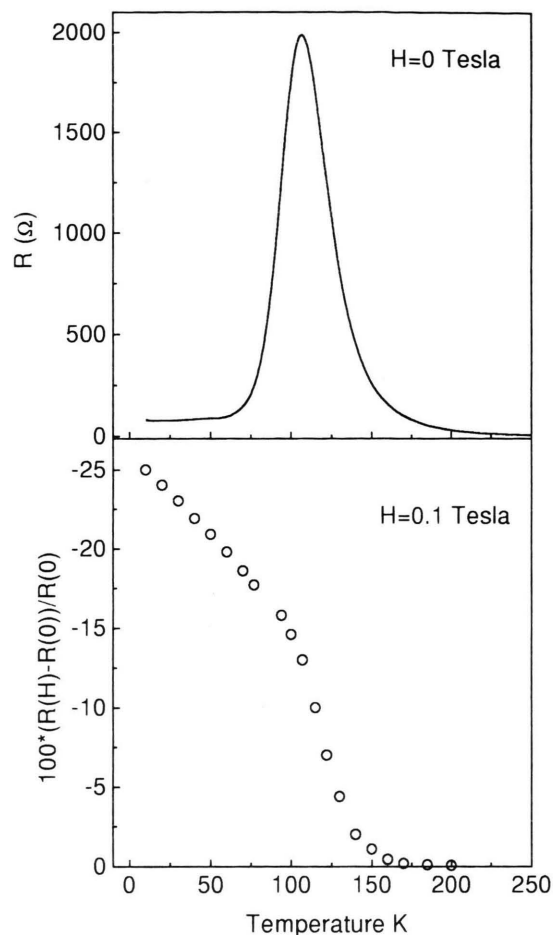


Fig. 3. The temperature dependences of the resistance R without field and the magnetoresistance MR with 0.1 Tesla field for $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$.

peaks are indexed with respect to the $\text{Sr}_3\text{Ti}_2\text{O}_7$ -type structure. The space group is $I4/mmm$ with the lattice parameters of the tetragonal unit cell being $a = 0.3884$ nm and $c = 1.9272$ nm. Good agreement between the calculated and observed positions and intensities of the lines indicates that this sample is a single phase of the $\text{Sr}_3\text{Ti}_2\text{O}_7$ -type, not a mixture with the ABO_3 and K_2NiF_4 -type structure.

The temperature dependence of the resistance R without field for $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ with $x = 0.2$ is depicted in the top panel of Figure 3. As is evident that the $R - T$ profile shows a sharp cusp at ~ 107 K (T_p), with semiconductor behavior above and metallic behavior below this temperature. Within the semiconductor range, the resistance of $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ falls steadily with increasing temperature, following

the law $R \sim \exp(T_0/T)^\nu$ with $\nu \simeq 1/4$. $\nu = 1/4$ corresponds to Mott's expression for variable-range hopping of charge carriers in a band of localized states in the absence of electron correlations [14].

The bottom panel of Fig. 3 shows the temperature dependence of the MR under an applied field of 0.1 Tesla. The MR at 10 K becomes as high as 25%. It is clear that the MR shows a gradual decline with increasing temperature below near T_p , followed by an abrupt decrease above this temperature. No MR peaks were observed. This kind of temperature profile of MR is reminiscent of that found in heteroepitaxial magnetic tunneling junctions containing magnetic electrodes of manganites separated by an insulating barrier layer [10]. Their temperature profiles of MR are very similar.

To understand this MR behavior it seems essential to consider the structure of the sample. As mentioned above, the layered perovskite crystals are a stack of ferromagnetic-metallic (FM) sheets composed of the MnO_2 bilayers which are separated by insulating $(\text{La}, \text{Ca})_2\text{O}_2$ layers and form a natural array of FM / Insulator / FM junctions. So do the polycrystals within a grain. Previous studies showed that at low temperatures ($T < T_c$) the moments of the MnO_2 bilayers are essentially parallel within domains separated by domain boundaries lying on the $(\text{La}, \text{Ca})_2\text{O}_2$ layers [12]. Recent experiments indicate that the low-temperature phase of this system may consist of mostly antiferromagnetic and some ferromagnetic static order between the adjacent MnO_2 bilayers [13, 15]. In this sense, a relatively low field can easily align the moment of the adjacent MnO_2 and makes it more easy that nearly fully spin-polarized carriers tunnel through the insulating $(\text{La}, \text{Ca})_2\text{O}_2$ layers between the adjacent MnO_2 bilayers, giving a large low-field tunneling MR which increases with decreasing temperature. At temperature ($T \geq T_c$), the magnetic ordering along the c -direction is destroyed. A low field (such as a 0.1 Tesla field) can not align the c -direction spins, so the MR in these temperature ranges are very small.

In Fig. 4, we display the normalized isothermal MR and Magnetization (M) as a function of the applied field at a temperature of 10 K. Resistance shows an abrupt decrease in the low-field range and becomes constant when the M is saturated under applied fields of about 0.3 Tesla. This confirms the conclusion that a relatively low field can easily align the moment of the adjacent MnO_2 , thus giving a large low-field tunneling MR at low temperature. As is evident, the

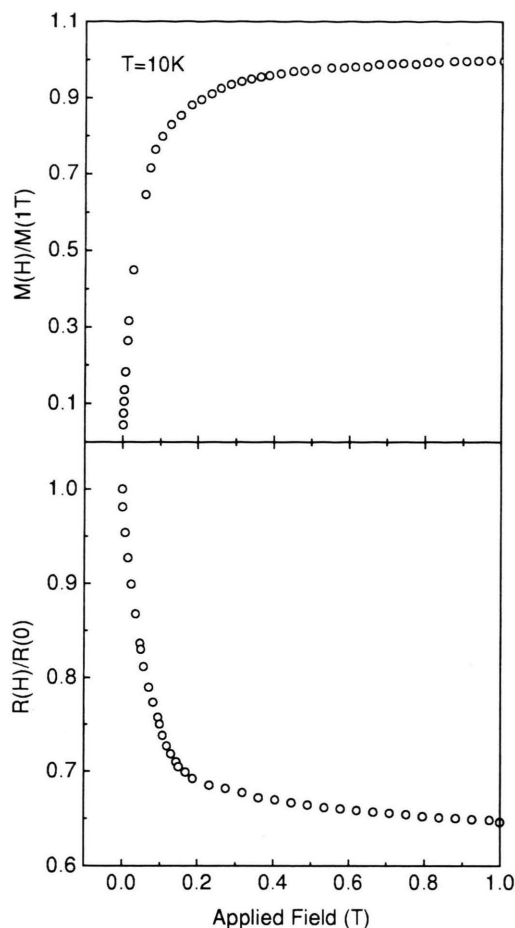


Fig. 4. The normalized isothermal MR and Magnetization (M) as a function of applied field at the temperature of 10 K.

sensitivity of the MR response to the field is high in the low-field range.

Because of the polycrystalline nature of the sample, the grain-boundary transport also plays a role in the low-field MR at low temperature. That is to say the spin-polarized tunneling between grains [11] may be another factor which causes the high low-field MR at low temperature in the layered $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ polycrystal, although currently its weight can not be known.

The low-field MR of polycrystalline $\text{La}_{1-x}\text{Ca}_x\text{MnO}_3$ ($x = 0.2$), the perovskite parent compound of two-layered $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ was also measured at 10 K for comparison: MR $\sim -4\%$, -6.5% and -8% under applied fields of 0.05, 0.1 and 0.15 Tesla, respectively. These values are significantly smaller than those (-17% , -25% and -30% under applied

fields of 0.05, 0.1 and 0.15 Tesla, respectively) for the two-layered $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$. The MR value of -30% under an applied of 0.15 Tesla is also larger than that ($\sim -12\%$ under 0.15 Tesla field at 4.2 K) observed in polycrystalline $\text{La}_{1-x}\text{Ca}_x\text{MnO}_3$ films [16]. However, the low-field MR values in two-layered $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ polycrystal are smaller than those found in heteroepitaxial magnetic tunneling junction containing magnetic electrodes of the manganites separated by an insulating barrier layer [10] and two-layered single crystals [12, 13]. This

may be the reason why much more impurities and defects are found in the former case, while no or few are found in the two latter cases. These impurities and defects can enhance the spin-flip effects of the tunneling carriers and reduce the low-field MR in polycrystals [17].

In short, by the sol-gel method a polycrystal bulk of two-layered $\text{La}_{2-2x}\text{Ca}_{1+2x}\text{Mn}_2\text{O}_7$ with $x = 0.2$ was successfully synthesized, and a high sensitive low-field TMR was discovered at low temperature.

This work is supported by NMS and NSFC.

- [1] M. N. Baibich, J. M. Broto, A. Fert, F. Nguyen Van Dau, F. Petroff, P. Etienne, G. Creuzet, A. Friederich, and J. Chazelas, *Phys. Rev. Lett.* **61**, 2472 (1988).
- [2] G. Binasch, P. Grünberg, F. Sauerbach, and W. Zinn, *Phys. Rev. B* **39**, 4828 (1989).
- [3] J. Q. Xiao, J. S. Jiang, and C. L. Chien, *Phys. Rev. Lett.* **68**, 3749 (1992).
- [4] A. E. Berkowitz, J. R. Mitchel, M. J. Carey, A. P. Young, S. Zhang, F. E. Spada, F. T. Parker, A. Hutten, and G. Thomas, *Phys. Rev. Lett.* **68**, 3745 (1992).
- [5] B. Dieny, V. S. Speriosu, S. S. P. Parkin, B. A. Gurney, D. R. Wilhoit, and D. Mauri, *Phys. Rev. B* **43**, 1279 (1991).
- [6] T. Miyazaki and N. Tezuka, *J. Magn. Magn. Mater.* **139**, L231 (1995).
- [7] J. S. Moodera, L. R. Kinder, T. M. Wong, and R. Meserve, *Phys. Rev. Lett.* **74**, 3237 (1995).
- [8] J. M. Coey, A. E. Berkowitz, L. Balcells, F. F. Putris, and A. Barry, *Phys. Rev. Lett.* **80**, 3815 (1998).
- [9] S. Lee, H. Y. Hwang, Boris I. Shraiman, W. D. Ratcliff II, and S-W. Cheong, *Phys. Rev. Lett.* **82**, 4508 (1999).
- [10] Y. Lu, X. W. Li, G. Q. Gong, G. Xiao, A. Gupta, P. Lecouer, J. Z. Sun, Y. Y. Wang, and V. P. Dravid, *Phys. Rev. B* **54**, R8357 (1996).
- [11] H. Y. Hwang, S-W. Cheong, N. P. Ong, and B. Batlogg, *Phys. Rev. Lett.* **77**, 2041 (1996).
- [12] T. Kimura, Y. Tomioka, H. Kuwahara, A. Asamitsu, M. Tamura, and Y. Tokura, *Science* **274**, 1698 (1996).
- [13] T. Kimura, A. Asamitsu, Y. Tomioka, and Y. Tokura, *Phys. Rev. Lett.* **79**, 3720 (1997).
- [14] N. F. Mott, *Adv. Phys.* **21**, 785 (1972).
- [15] T. G. Perring, G. Aeppli, T. Kimura, Y. Tokura, and M. A. Adams, *Phys. Rev. B* **58**, R14693 (1998).
- [16] X. W. Li, A. Gupta, G. Xiao, and G. Q. Gong, *Appl. Phys. Lett.* **71**, 1124 (1997).
- [17] P. Lyu, D. Y. Xing, and J. Dong, *Phys. Rev. B* **58**, 54 (1998).